



# Zentrum für Technomathematik

Fachbereich 3 – Mathematik und Informatik

## On the statistical distribution of elastic moduli polycrystals

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Report 10–02

Berichte aus der Technomathematik

Report 10–02

April 2010



# On the statistical distribution of elastic moduli in polycrystals\*

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file: DistElMod.V6.tex 14.01.2010

May 11, 2010

## Abstract

A method for calculating the distributions of the components of stiffness and compliance tensors of cubic polycrystalite material from parameters of single crystals is developed. Their mathematical form is derived under the assumption of normal distributions for the members of the rotation group  $SO(3)$ . Finally the impact of texture on the distributions of Young's modulus  $E$ , shear modulus  $G$  and Poisson's ratio  $\nu$  analyzed.

**1. The problem** The elastic properties of polycrystalite material are described by the compliance tensor  $S_{ijkl}$  or the stiffness tensor  $C_{ijkl}$  ( $i, j, k, l = 1, 2, 3$ ) [1, 2, 3]. Because of the symmetry of strain and stress tensor, and for energetical considerations both tensors can be expressed by symmetric matrices  $S_{ij}$  and  $C_{ij}$  ( $i, j = 1, 2, \dots, 6$ ), respectively, each containing 21 independent components [1]. In the sequel we focus on the compliance tensor, the components of which are termed elastic moduli.

For anisotropic crystals the symmetry properties of the various crystal systems reduce the number of independent matrix elements. As an example, there are only three independent elements for a cubic crystal system [2, 3].

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\*Paper prepared in the framework of the Sonderforschungsbereich 747 "Mikrokalturnformen - Prozesse, Charakterisierung, Optimierung", project B2 "Verteilungsbasierte Simulation", University of Bremen, supported by Deutsche Forschungsgemeinschaft

To describe a single crystal, an orthogonal coordinate system tightly connected to the crystal axes may be chosen. Such a system is called a crystal physical coordinate system. We choose the directions of the cube edges as axes for a cubic crystal system [3]. In this coordinate system the compliance tensor  $s_{ijkl}$  may be represented in matrix form as

$$\begin{pmatrix} s_{11} & s_{12} & s_{12} & 0 & 0 & 0 \\ & s_{11} & s_{12} & 0 & 0 & 0 \\ & & s_{11} & 0 & 0 & 0 \\ & & & s_{44} & 0 & 0 \\ & & & & s_{44} & 0 \\ & & & & & s_{44} \end{pmatrix} \quad (1)$$

where  $s_{11} = s_{1111}$ ,  $s_{12} = s_{1122}$  and  $s_{44} = s_{1212}$ .

Now we define an orthogonal coordinate system in the specimen. This coordinate system may be chosen arbitrarily. We use the RTN system [3], which is commonly used for sheets and foils.

We describe the orientation of a crystalite in a polycrystal by a rotation  $g$ , which maps the probenfest system on the crystal physical system. These rotations constitute the rotation group  $\text{SO}(3)$ . For the tensor  $S_{ijkl}$  in the RTN system we have the tensor representation of the rotation group<sup>1</sup>:

$$S_{ijkl} = g_{ip}g_{jq}g_{km}g_{ln}s_{pqmn} \quad (2)$$

where  $g_{ij}$  denotes the components of the rotation matrix.

As crystalites in the polycrystal are randomly orientated, also the components  $g_{ij}$  show a random variation, which induces random variation in the values of the elastic moduli. If the orientations of the crystalite are uniformly distributed, the corresponding polycrystal is quasi isotropic. If a principal orientation exists, we have a texture.

This contribution considers the distributions of the matrix elements of the compliance tensor for a polycrystal that are induced by normal measures on  $\text{SO}(3)$ .

**2. Distributions on  $\text{SO}(3)$**  The general theory for distributions on locally compact groups is dealt with in [6, 7]. We consider distributions  $d\mu = f(g)dg$ ,  $g \in \text{SO}(3)$ , where  $dg$  is an invariant measure on  $\text{SO}(3)$ . If Rotation  $g$  is parametrized by Euler angles  $g = g(\varphi_1, \theta, \varphi_2)$ ,  $\theta \in [0, \pi]$  and  $\varphi_1, \varphi_2 \in [-\pi, \pi)$  we have [4, 8]

$$dg = \frac{\sin \theta d\theta}{2} \frac{d\varphi_1}{2\pi} \frac{d\varphi_2}{2\pi}$$

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<sup>1</sup>We sum over repeatedly occurring indices.

The texture function  $f(g)$  [5], which depends on the orientation, may be expanded as a series of generalized spheric functions

$$f(g) = \sum_{l=0}^{\infty} \sum_{m,n=-l}^l C_{mn}^l T_{mn}^l(g)$$

If  $f(g) = 1$ , the polycrystal is quasi isotropic. For this case, the distribution of the elastic moduli of hexagonal polycrystals has been approximated by quadratic polynomials in [9].

An important distribution class on  $SO(3)$  is the class of normal distributions. We define these according to [10, 12] in the following way: a measure  $\mu$  on  $SO(3)$  has a normal distribution, if  $\mu$  is infinitely divisible and not idempotent, and if we have for each irreducible representation  $T_g$  of the group that

$$\int_G T_g d\mu(g) = \exp \left[ \sum_{i,j} \alpha_{ij} A_i A_j + \sum_i \alpha_i A_i \right]$$

holds, where  $A_i$  are infinitesimal operators of this representation,  $(\alpha_{ij})$  is a positive definite symmetric matrix, and  $\alpha_i$  are real numbers. If  $\alpha_{ij} = 0$  for  $i \neq j$ , we have a canonical normal distribution [11] with analytical probability distribution

$$f(g) = \sum_{l=0}^{\infty} (2l+1) \exp[-l(l+1)p^2] \times \times \sum_{m=-l}^l \exp[m^2(q^2 - r^2)] \exp[-im(\varphi_1 + \varphi_2)] P_{mm}^l(\cos \theta) \quad (3)$$

Here,  $P_{mm}^l(x)$  are Jacobian polynomials with parameters  $q, r, p$ . However, though we consider orientation distributions belonging to the family of canonical distributions, eq. (3) is not useful for our purpose.

Another way to derive the desired distributions lies in the application of Parthasarathy's [12] central limit theorem for the rotation group. We need some additional notation for this theorem. Let  $g_n = \int_{SO(3)} g d\mu_n(g)$  be the mean of the measure  $\mu_n(g)$ ,  $n = 1, 2, 3, \dots$ ,  $\mu_n^{*n}(d\mu_2^{*2}(g) = \int_{SO(3)} \mu_2(gg_1^{-1}) d\mu_2(g_1))$   $n$ -fold convolution  $\mu_n$  and  $\mu_n(SO(3) \setminus U_e)$  the value of the measure  $\mu_n(g)$  outside the neighbourhood  $U_e$  of the null element  $e$  of the group.

A Central Limit Theorem for  $SO(3)$  (Parthasarathy). Let  $\{\mu_n\}$  ( $n=1,2,3,\dots$ ) be a sequence of distributions on  $SO(3)$ , which converges to a degenerate distribution in  $e$  as  $n \rightarrow \infty$ . The sequence  $\mu_n^{*n}$  of convolutions converges to a normal distribution if and only if for  $n \rightarrow \infty$  holds

1.  $n(1 - |g_n|) < \infty$

2.  $n\mu_n(SO(3)\setminus U_e) \rightarrow 0$

With the additional condition  $\lim_{n \rightarrow \infty} n(e - g_n) = A$  the parameters  $\alpha_{ij}$  and  $\alpha_i$  of the distribution  $\mu_n^{*n}$  are given by

$$-A = \sum_{i=1}^3 \sum_{j=1}^3 \alpha_{ij} a_i a_j + \sum_{i=1}^3 \alpha_i a_i$$

with  $a_i = \lim_{t \rightarrow 0} \frac{g_i(t) - e}{t}$ , where  $g_i(t)$  are the one-parametric subgroups of  $SO(3)$ .

**3. Simulation of orientation distributions** Due to the CLT we may specify an order  $\mu_n(g)$ . We use the concept of small rotations [13] for this purpose. The set of small rotations corresponds to a certain set of Euler angles:

$$\begin{cases} 1 - (e'_z, e_z) = 1 - \cos \theta \leq a \\ 1 - (e'_x, e_x) = 1 - \cos \varphi_1 \cos \varphi_2 + \sin \varphi_1 \sin \varphi_2 \cos \theta \leq b \\ 1 - (e'_y, e_y) = 1 + \sin \varphi_1 \sin \varphi_2 - \cos \varphi_1 \cos \varphi_2 \cos \theta \leq b \\ 0 \leq a, b \ll 1 \end{cases} \quad (4)$$

Here,  $(e_x, e_y, e_z)$  form a basis before a small rotation and  $(e'_x, e'_y, e'_z)$  thereafter. Given that  $a, b \ll 1$ , (4) simplifies to

$$\frac{\theta^2}{2} \leq a \quad , \quad \frac{(\varphi_1 + \varphi_2)^2}{2} \leq b$$

The set of small rotations now is represented by the region

$$\Pi(a, b) = \left\{ (\varphi_1, \theta, \varphi_2) : \theta \leq \sqrt{2a} = \bar{a} \quad , \quad |\varphi_1 + \varphi_2| \leq \sqrt{2b} = \bar{b} \right\}$$

The sequence of measures  $d\mu_n = f_n(g)dg$ , with  $dg$  denoting the invariant measure,  $f_n$  the rectangular distribution in  $\Pi(a_n, b_n)$ ,  $a_n = \bar{a}/\sqrt{n}$ ,  $b_n = \bar{b}/\sqrt{n}$  and

$$f_n(\varphi_1, \theta, \varphi_2) = \begin{cases} \frac{2}{1 - \cos a_n} \cdot \frac{4\pi^2}{b_n(4\pi - b_n)} & (\varphi_1, \theta, \varphi_2) \in \Pi(a_n, b_n) \\ 0 & (\varphi_1, \theta, \varphi_2) \notin \Pi(a_n, b_n) \end{cases} \quad (5)$$

corresponds to the sequence of convolutions

$$d\mu_n^{*n} = \left[ \int_{SO(3)} dg_{n-1} f_n(gg_{n-1}^{-1}) \dots \int_{SO(3)} dg_1 f_n(g_2 g_1^{-1}) f_n(g_1) \right] dg \quad , \quad (6)$$

which converges for  $t \rightarrow \infty$  to the canonical normal distribution on  $\text{SO}(3)$  with parameters  $q^2 = \bar{a}^2/8$  and  $r^2 = \bar{b}^2/6$  [13].

The realisation of the random variable  $g \in \text{SO}(3)$  with distribution 6) is the product of the small random rotations  $g = g_1 g_2 \dots g_n$ , where  $g_i = g(\varphi_1^i, \theta^i, \varphi_2^i) \in \text{SO}(3)$  with density (5). The Euler angles  $(\varphi_1^i, \theta^i, \varphi_2^i)$  are given by [13]:

$$\varphi_1^i = \begin{cases} -(\pi + b_n) + \sqrt{b_n^2 + 2b_n(4\pi - b_n)\xi_1^i}; & \xi_1^i < \frac{3b_n}{2(4\pi - b_n)} \\ \left(\frac{b_n}{4} - \pi\right) + \frac{4\pi - b_n}{2}\xi_1^i; & \frac{3b_n}{2(4\pi - b_n)} \leq \xi_1^i < 1 - \frac{3b_n}{2(4\pi - b_n)} \\ (\pi + b_n) - \sqrt{b_n^2 + 2b_n(4\pi - b_n)(1 - \xi_1^i)}; & 1 - \frac{3b_n}{2(4\pi - b_n)} \leq \xi_1^i \end{cases}$$

$$\theta^i = \arccos(1 - \xi_2^i(1 - \cos a_n)) \quad (7)$$

$$\varphi_2^i = \begin{cases} \pi - \xi_3^i(b_n + \pi + \varphi_1^i); & -\pi \leq \varphi_1^i < b_n - \pi \\ -\varphi_1^i + 2b_n(\xi_3^i - \frac{1}{2}); & |\varphi_1^i| \leq \pi - b_n \\ -\pi + \xi_3^i(b_n + \pi - \varphi_1^i); & \pi - b_n < \varphi_1^i \leq \pi \end{cases}$$

where  $\xi_1^i, \xi_2^i, \xi_3^i$  are independent and uniformly distributed in  $[0,1]$ .

In [13] it was shown that convolution parameters  $n \geq 20$  generate a good agreement of  $\mu_n^{*n}$  with a normal distribution with density given in (3).

The distribution of crystalite orientations may be modelled by Monte Carlo simulation and the material parameters of interest are subsequently derived from these. Figure 1 shows the distribution of the projection of the nutation angle  $\theta$  on the sphere for selected values of  $\bar{a} = 0.2$ ,  $\bar{b} = 0.5$  and  $n$ . This distribution results from 10000 realizations. If the polycrystal material has a certain texture  $g_0$  with main orientation different from the cube coordinate edges, the resulting normal distribution (in the sense of  $\text{SO}(3)$ ) of the nutation angle is shifted towards this actual orientation.

**4. Results.** Equation (2) leads to the tensor (1)

$$\begin{aligned}
S_{11} &= s_{11} - 2\mu^s (g_{11}^2 g_{21}^2 + g_{11}^2 g_{31}^2 + g_{21}^2 g_{31}^2) \\
S_{12} &= s_{12} + \mu^s (g_{11}^2 g_{12}^2 + g_{21}^2 g_{22}^2 + g_{31}^2 g_{32}^2) \\
S_{44} &= s_{44} + 4\mu^s (g_{12}^2 g_{13}^2 + g_{22}^2 g_{23}^2 + g_{32}^2 g_{33}^2) \\
S_{14} &= 2\mu^s (g_{11}^2 g_{12} g_{13} + g_{21}^2 g_{22} g_{23} + g_{31}^2 g_{32} g_{33}) \\
S_{16} &= 2\mu^s (g_{11}^3 g_{12} + g_{21}^3 g_{22} + g_{31}^3 g_{32})
\end{aligned} \tag{8}$$

where  $\mu^s = s_{11} - s_{12} - s_{44}/2$  is the measure of anisotropy<sup>2</sup>. [14]. If the elements of the rotation matrix  $g_{ij}$  is parametrized by Euler angles, the equations (8) may be given as

$$S_{ij} = s_{ij} + \mu^s C_{ij} \Lambda_{ij}(\varphi_1, \theta, \varphi_2)$$

Here, we have a normal distribution of  $g = g(\varphi_1, \theta, \varphi_2)$  on SO(3) around a preferred orientation  $g_0 = g(\varphi_1^0, \theta^0, \varphi_2^0)$ .

We consider aluminium as example of a polycrystal material with  $s_{11} = 1.57$ ,  $s_{12} = -0.57$  and  $s_{44} = 3.51$  (all units:  $10^{-11}$  GPa<sup>-1</sup>) [15] and a rolling texture (1,1,2)[ $\bar{1}, \bar{1}, 1$ ] [14, 16]. The principal representation matrix for this orientation is

$$\begin{pmatrix} -\frac{1}{\sqrt{3}} & \frac{1}{\sqrt{2}} & \frac{1}{\sqrt{6}} \\ -\frac{1}{\sqrt{3}} & -\frac{1}{\sqrt{2}} & \frac{1}{\sqrt{6}} \\ \frac{1}{\sqrt{3}} & 0 & \frac{\sqrt{2}}{\sqrt{3}} \end{pmatrix} \tag{9}$$

The parameters  $\bar{a}$  and  $\bar{b}$  define the strength of the texture: smaller parameters indicate crystallites lying closer to the main orientation. Figures 2 and 3 allow comparing the dispersion of the nutation angle corresponding to various  $\bar{a}$  and  $\bar{b}$  for the same given RTN texture. Obviously the dispersion of  $\Lambda_{ij}$  and  $S_{ij}$  depend on the strength of the texture and increase with its decrease.

For further insight into the relation between  $\bar{a}$  and  $\bar{b}$  and the corresponding distributions of the various  $S_{ij}$  quantities we did Monte Carlo simulations

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<sup>2</sup>All other  $S_{ij}$  result from cyclical re-arrangement of indices, e.g.

$$S_{66} = s_{44} + 4\mu^s (g_{11}^2 g_{12}^2 + g_{21}^2 g_{22}^2 + g_{31}^2 g_{32}^2)$$

with 10000 realizations for several values of  $(\bar{a}, \bar{b})$ . The resulting  $S_{ij}$  distributions are shown in Figures 4 - 11.

Obviously, with increasing parameters  $\bar{a}$  and  $\bar{b}$  the set of small rotations  $\Pi(a, b)$  increases to the full set  $\{0 \leq \theta \leq \pi; -\pi \leq \varphi_1; \varphi_2 \leq \pi\}$  and the distribution of orientations becomes a uniform distribution on  $SO(3)$ .

Using the distributions of the elastic moduli we are able to derive the distributions of the Young's modulus in arbitrary directions  $E_{\alpha\alpha}$ , of the Shear modulus  $G_{\alpha\alpha}$  or of Poisson's ratio  $\nu_{\alpha\alpha}$  [14]. As an example we have

$$E_{WW} = 1/S_{11} \quad , \quad G_{WQ} = 1/S_{44} \quad , \quad \nu_{WQ} = -S_{12}/S_{11}$$

Figures 12-17 display parameter distributions under two different textures, each obtained for  $\bar{a} = 0.15$ ,  $\bar{b} = 0.3$  and  $n = 100$ , based on 10000 Monte Carlo realizations. Figures in the left column contain  $E_{WW}$ ,  $G_{WQ}$  and  $\nu_{WQ}$  for the texture  $(1,1,2)[\bar{1}, \bar{1}, 1]$  from above, while the right column of figures refers to the texture  $(1,1,0)[1, \bar{1}, 2]$ . The Young's modulus in arbitrary directions  $\alpha$  in the rolling plane can be calculated according to the formulae

$$S_{\alpha\alpha} = S_{11} \cos^4 \alpha + (2S_{12} + S_{66}) \cos^2 \alpha \sin^2 \alpha + S_{22} \sin^4 \alpha$$

and  $E_{\alpha\alpha} = 1/S_{\alpha\alpha}$ . Figure 18 shows the dependence of mean and quantiles of the Young's modulus distribution on the angle  $\alpha$  between the rolling direction and direction concerned for the texture  $(1,1,2)[\bar{1}, \bar{1}, 1]$ . The dependence was obtained for  $\bar{a} = 0.2$ ,  $\bar{b} = 0.3$  and  $n = 100$ , based on 2000 Monte Carlo realizations.

Having determined the distributions of interest by Monte Carlo simulation we can derive descriptive quantities for this distribution. As an example, we have for the Shear modulus in the left column of Figures 12-17 (all values in GPa)

Mean	SD	Min	Q1	Median	Q3	Max
26.54	0.40	25.08	26.27	26.54	26.81	27.87

Figure 19 displays the distribution of  $E_{QQ} = 1/S_{22}$ , again for the texture  $(1,1,2)[\bar{1}, \bar{1}, 1]$ .

**5. Discussion.** In the preceding sections we showed a method for deriving the distribution of the elastic moduli  $S_{ij}$  assuming a normal distribution for the orientation of single crystalites in a textured polycrystal. As the  $S_{ij}$  are complicated functions of the Euler angles, we cannot describe their distributions by simple standard distributions. However, we can obtain all characterizing quantities of these distributions up to a histogram or density

estimate of the complete distribution in a relatively simple way. As additionally all distributions have compact support, a first approximation by polynomials or mixtures of these may be considered. For strongly textured material ( $\bar{a} \leq 0.2$  and  $\bar{b} \leq 0.3$ ) the distributions of some  $S_{ij}$  may well be approximated by normal or log-normal distributions. A more precise description of the  $S_{ij}$  distributions can be obtained by mixtures of distribution. This aspect, however, lies outside the scope of this contribution.

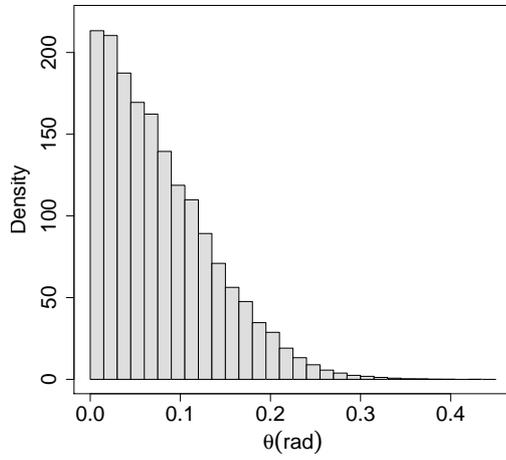
A further problem is the determination of the parameters  $q$  and  $r$  (corresponding to  $\bar{a}$  and  $\bar{b}$ ) of the canonical normal distribution (3) of the experimental material. These can be derived from the pole figures [11].

All distributions presented here were obtained while neglecting possible correlations between properties of different crystalites. Including such correlations into the consideration, which is especially necessary for analyzing multiphase polycrystals, will be a further step.

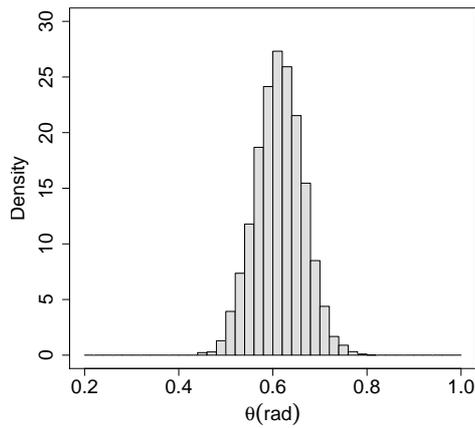
The method employed here cannot only be used for the calculation of elastic parameters, but also for the calculation of further material parameters that allow representation by tensor quantities.

## References

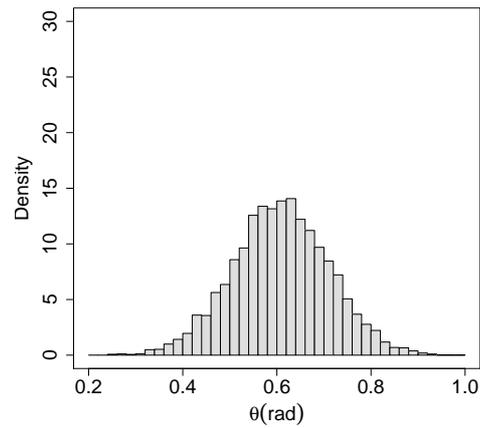
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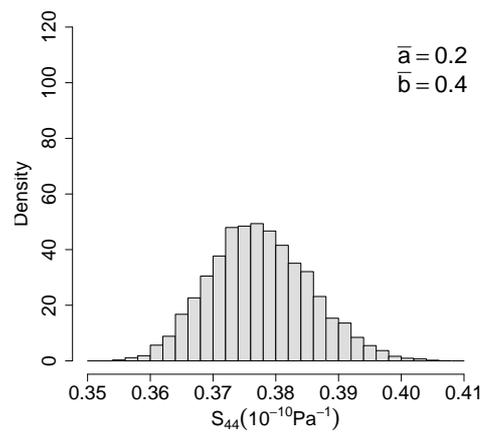
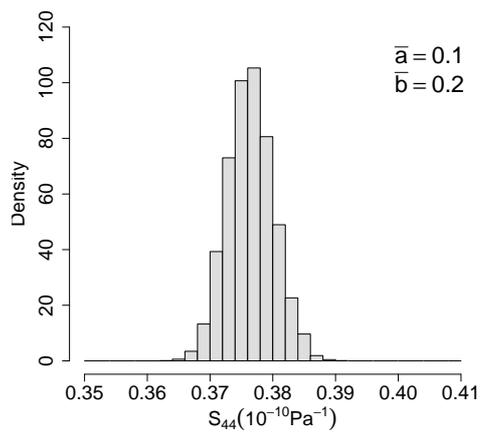
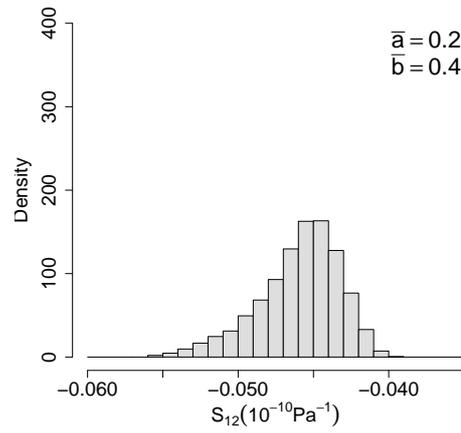
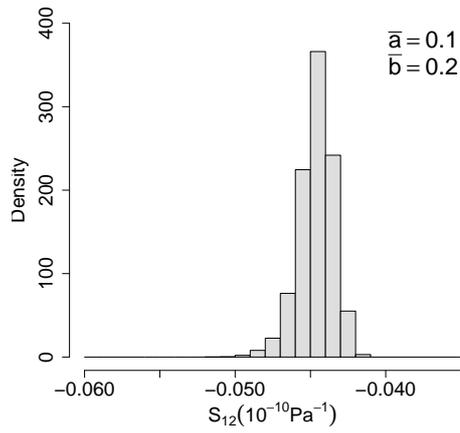
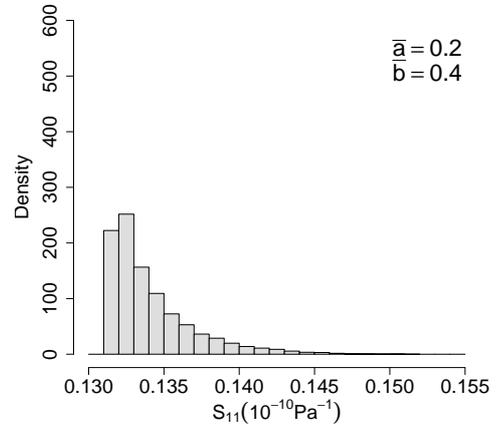
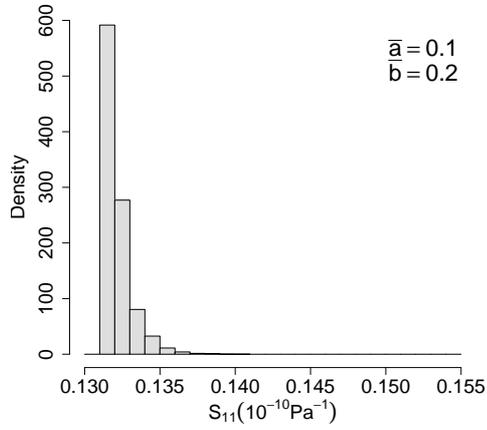
**Figure 1.** Histogram of the distribution of the mutation angle  $\theta$  for  $\bar{a} = 0.2$ ,  $\bar{b} = 0.5$  and  $n = 50$

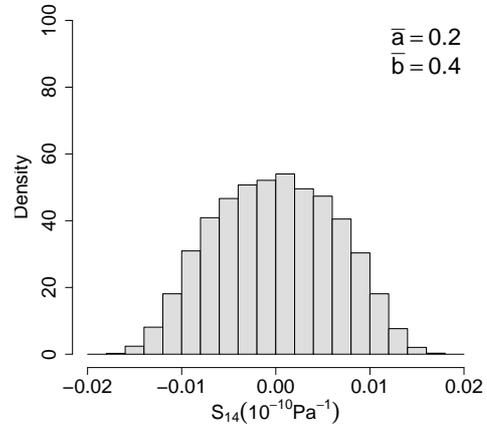
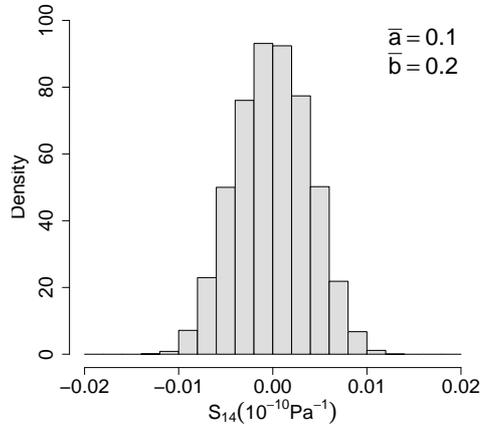


**Figure 2.** Histogram of the distribution of the mutation angle  $\theta$  for  $\bar{a} = 0.1$ ,  $\bar{b} = 0.2$  and  $n = 100$ , given a RTN texture

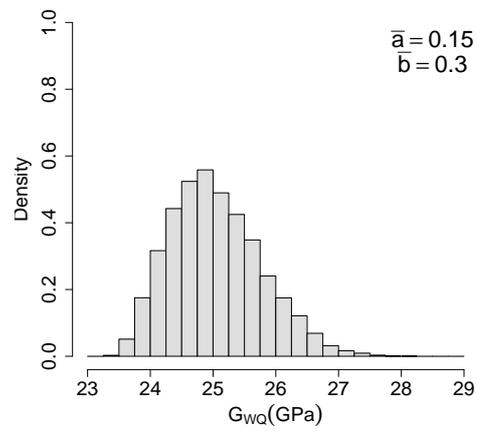
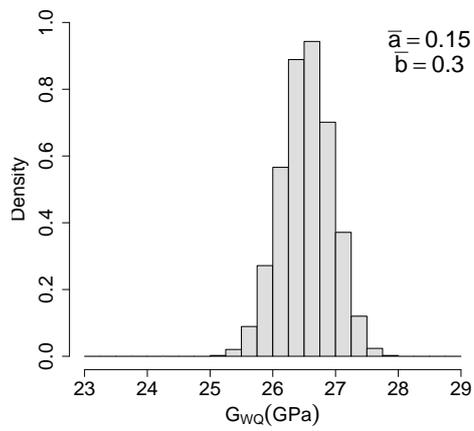
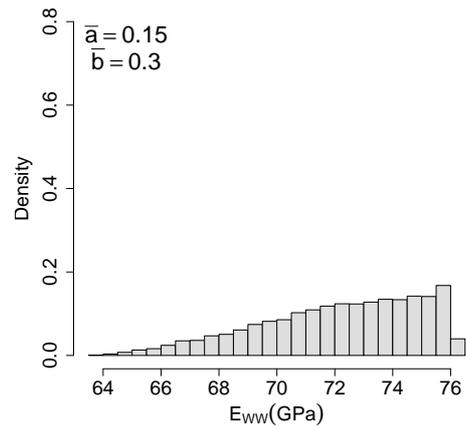
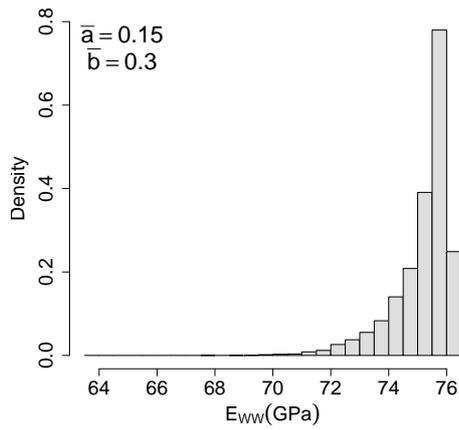


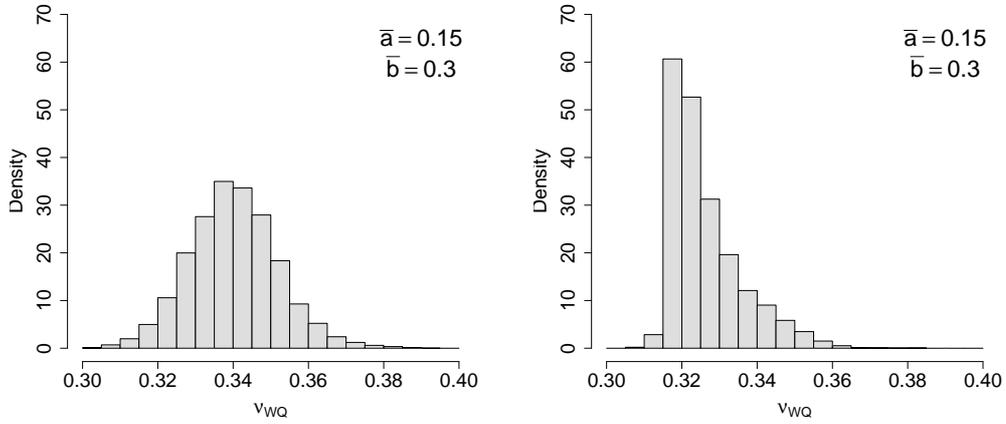
**Figure 3.** Histogram of the distribution of the mutation angle  $\theta$  for  $\bar{a} = 0.2$ ,  $\bar{b} = 0.4$  and  $n = 100$ , given a RTN texture



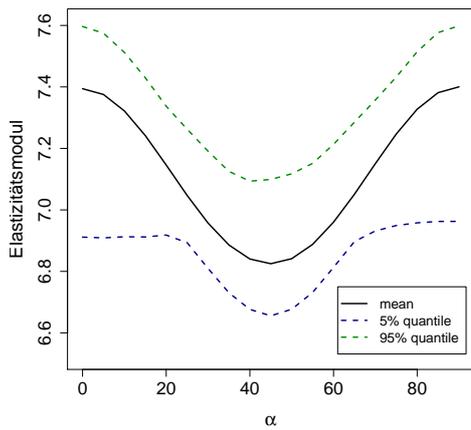


**Figures 4-11.** Distributions of various  $S_{ij}$  for a set of  $\bar{a}$  und  $\bar{b}$  values.

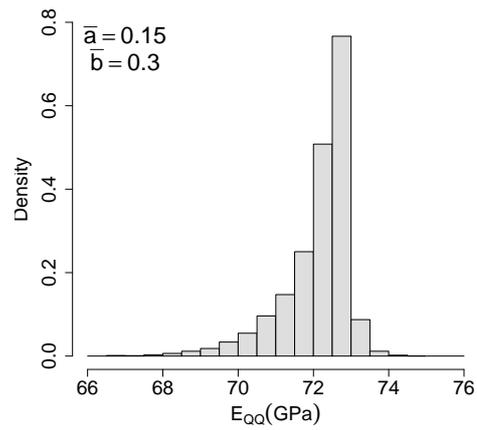




**Figure 12-17.** Histogram of the distribution of various elastic parameters for  $\bar{a} = 0.15$ ,  $\bar{b} = 0.3$  and  $n = 100$ . Left column: texture  $(1,1,2)[\bar{1},\bar{1},1]$ , right column: texture  $(1,1,0)[1,\bar{1},2]$



**Figure 18.** Diagram of the Young's modulus  $E_{\alpha\alpha} = 1/S_{\alpha\alpha}$  for  $\bar{a} = 0.2$ ,  $\bar{b} = 0.3$  and  $n = 100$  with the quantiles;  $\alpha \in [0, \pi/2]$



**Figure 19.** Histogram of the distribution of the Young's modulus  $E_{QQ} = 1/S_{22}$  for  $\bar{a} = 0.15$ ,  $\bar{b} = 0.3$  and  $n = 100$